Sustainable Chemistry & Engineering

pubs.acs.org/journal/ascecg

Research Article

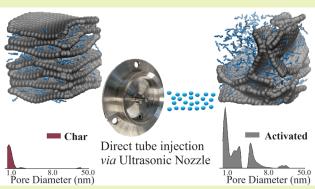
¹ Ultrasonic Spray Nozzle-Mediated Green Activation for Hierarchical ² Pore-Structured Carbon Beads

3 Michael Hough, Artur Deditius, Neil Robinson, Gerd Elmar Schröder-Turk,

⁴ Jacob Judas Kain Kirkensgaard, Volodymyr M. Gun'ko, Alexander V. Neimark, Katsumi Kaneko, ⁵ and Piotr Kowalczyk*



10 fabricating carbon beads with customizable ultra-, super-, and 11 mesopores. This study showcases noteworthy results from 12 subjecting spherical char particles to activation processes involving 13 carbon dioxide, a mixture of carbon dioxide and micron-sized water 14 droplets, and highly concentrated supercritical steam at a temper-15 ature of 1173 K for durations of 3 and 5 h. Through pulse-field 16 gradient nuclear magnetic resonance measurements, it was noted 17 that carbon beads produced using USN-generated highly con-18 centrated supercritical steam displayed remarkably elevated intra-



¹⁹ bead self-diffusivity of *n*-hexane. Inductively coupled plasma-optical emission spectroscopy demonstrates superior gold recovery ²⁰ kinetics from cyanide solutions compared to that from an industrial benchmark. The energy expenditure for USN-generated steam, ²¹ producing carbon beads with an apparent surface area of 2691 m²/g, is estimated at 97 J per 1 m² of carbon. This contrasts with the ²² traditional steam generation method requiring approximately the energy of 190 J/m² for activated carbon with an S_{BET} of 2130 m²/g, ²³ making the USN-assisted activation method a more environmentally friendly and sustainable option with nearly half the energy ²⁴ consumption.

25 **KEYWORDS:** Energy management, ultrasonic spray nozzle, steam activation, hierarchically porous carbon beads, 26 pulse-field gradient nuclear magnetic resonance, small-angle X-ray scattering

27 INTRODUCTION

28 Nanoporous carbons have been widely and predominantly 29 applied to green technologies.¹⁻¹⁰ One of the critical challenges 30 in the production of nanoporous carbons is to achieve high-31 quality nanoporous carbons while minimizing energy con-32 sumption. The traditional energy-intensive activation methods 33 do not align well with green technology.^{11,12} It is necessary to 34 innovate the production of nanoporous carbons having 35 customized characteristics with less energy. One promising 36 approach to achieve this is by leveraging the capabilities of an 37 ultrasonic spray nozzle (USN), which has the potential to 38 produce high-quality carbons while reducing energy consump-39 tion.

⁴⁰ USN technology is an emerging technology that has found ⁴¹ extensive applications in various fields, including spray coating, ⁴² fabrication of nanoparticles, and synthesis of functional ⁴³ nanomaterials and catalysts.^{13–18} A USN is a low-power ⁴⁴ consumption device that operates by converting high-frequency ⁴⁵ standing waves into mechanical energy, which is then transferred ⁴⁶ into a liquid. USN technology has the potential to accelerate and precisely control the activation process of char. This is achieved 47 by generating highly concentrated supercritical steam with 48 uniform micrometer-sized droplets, which are sprayed directly 49 into a quartz tube reactor without the need for a carrier gas. This 50 innovative approach can lead to the production of high-quality 51 nanoporous carbons. The use of USN-aided activation offers 52 significant advantages over traditional char activation that relies 53 on a water-heating process conducted outside a reactor to 54 produce steam. The steam is subsequently combined with a 55 carrier gas through a bubbling process and transported to the 56 reactor.¹¹ The need to sustain high temperatures for the steam 57 entails additional energy consumption to operate ribbon heaters 58

Received:July 10, 2023Revised:November 23, 2023Accepted:December 6, 2023



A

Table 1. Activation Parameters Used for the Production of HPSCBs^a

Sample	Activation method and parameters	Burn-off (wt %)	Yield (wt %)
HPSCB-CO ₂ -3h	CO ₂ , 1173 K for 3 hours, ~2.0 dm ³ /min	19 ± 0.7	15 ± 0.11
HPSCB-CO ₂ /H ₂ O- 3h	Ultrasonically generated mist of uniform 18 μm water droplets sprayed to CO ₂ (~2.0 dm ³ /min), 1173 K, for 3 hours	31 ± 3.4	12 ± 0.10
HPSCB-H ₂ O-3h	Ultrasonically generated mist of uniform 18 μ m water droplets sprayed to the quartz tube reactor, 1173 K, for 3 hours	48 ± 2.5	9 ± 0.08
HPSCB-CO ₂ -5h	CO ₂ , 1173 K for 5 hours, ~2.0 dm ³ /min	24 ± 1.6	14 ± 0.06
HPSCB-CO ₂ /H ₂ O- 5h	Ultrasonically generated mist of uniform 18 μm water droplets sprayed to CO $_2$ (~2.0 dm³/min), 1173 K, for 5 hours	47 ± 2.7	9 ± 0.07
HPSCB-H ₂ O-5h	Ultrasonically generated mist of uniform 18 μ m water droplets sprayed to the quartz tube reactor, 1173 K, for 5 hours	79 ± 10.9	4 ± 0.09

^aThe burn-off was determined using the sample weights before and after activation. The production yield was calculated by comparing the weight of the dry precursor sample to the weight of the dry carbon sample. The error in the burn-off degree and production yield is calculated as the standard deviation derived from three repeated experiments.

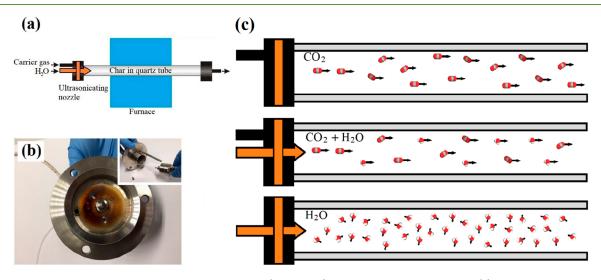


Figure 1. Scheme of the ultrasonic spray nozzle-quartz tube reactor (USN-QTR) used for the activation processes (a). An ultrasonic cone-shaped 120 kHz spray nozzle is incorporated into a stainless-steel sealing flange of the furnace; a nylon baffle distributes the carrier gas flow over the ultrasonic spray nozzle (b). Panel (c) provides a schematic diagram illustrating the activation of char using three different methods: CO_2 activation (top panel), CO_2 with ultrasonically sprayed micrometer-sized water droplet activation (middle panel), and highly concentrated supercritical steam activation without the use of a carrier gas (bottom panel).

⁵⁹ or heating tapes. Moreover, the utilization of USN-aided ⁶⁰ activation is suitable for producing nanoporous carbons with ⁶¹ customized pore structures, including hierarchical nanoporous ⁶² activated carbons.^{19,20} As far as we are aware, the integration of a ⁶³ USN into char activation processes has not been explored in ⁶⁴ previous research.

Hierarchical pore-structured carbon beads produced from 65 66 ion-exchange resin beads possess numerous advantageous 67 characteristics that make them highly suitable for a wide range 68 of applications. These advantages include high level of chemical purity, good fluidity and packaging, a smooth outer surface 69 70 topology, high mechanical strength, resistance to abrasion and 71 dust formation, and the ability to customize surface area and 72 porosity to meet specific requirments.^{20,21} These applications 73 encompass a broad range of uses and are not restricted to extracorporeal detoxification, hemo- and enterosorption, 74 alleviating chemotherapy side effects, extracting precious metals 75 from low-grade ore through adsorption from cyanide solutions, 76 and catalyst support.^{20–27} To ensure both high chemical purity 77 and precise control over the hierarchical pore structure of 78 79 carbon beads, it is crucial to develop and construct innovative 80 reactors for char activation that eliminate the requirement for 81 external chemical activating agents. To improve both energy and

water efficiency in the activation process that utilizes steam, it is s2 imperative to reduce the costs associated with steam production s3 and its transportation. This can be achieved by adopting state-ofthe-art USN technology, which offers a promising solution for s5 cost reduction without compromising the steam utilization s6 performance. 87

In this study, we developed and built an ultrasonic spray 88 nozzle-quartz tube reactor (USN-QTR) for char activation. We 89 applied the USN-QTR to produce two series of hierarchical 90 pore-structured carbon beads (HPSCBs) through activation 91 processes lasting 3 and 5 h. These processes included CO₂ 92 activation, ultrasonically generated micron-sized water droplets 93 injected with CO₂ activation, and highly concentrated super- 94 critical steam activation without the use of a carrier gas (Table 1 95 ti and Figure 1). The impact of alterations in accessible and 96 fl inaccessible pore volume and surface area fractions, pore size 97 distributions, and intrabead self-diffusivity of n-hexane is 98 examined through high-resolution nitrogen adsorption, small- 99 and wide-angle X-ray scattering, and pulse-field gradient nuclear 100 magnetic resonance spectroscopy. In the context of an 101 industrially relevant problem, we have examined the impact of 102 USN-assisted activation on the kinetics of gold dicyanide 103 recovery. This investigation was conducted using inductively 104

¹⁰⁵ coupled plasma-optical emission spectroscopy (ICP-OES). The ¹⁰⁶ energy cost of steam generation using a USN and water heating ¹⁰⁷ for the production of 1 m^2 of the apparent BET surface area has ¹⁰⁸ been estimated and thoroughly discussed.

109 EXPERIMENTAL SECTION

Synthesis. HPSCBs were prepared through a two-step activation 111 process, with carbonization performed as a separate step followed by 112 activation. The carbon precursor employed consisted of beads of an 113 acidic, macroporous cation exchange resin synthesized from a styrene-114 divinylbenzene copolymer (mean bead size = $670 \pm 50 \ \mu\text{m}$, S_{BET} = 15 115 m²/g, V_p = 0.144 cm³/g, and an average pore size D = 32.72 nm) and 116 supplied by Lewatit MonoPlus SP 112 H, Lanxess Pty Ltd. (Germany). 117 High-purity N₂ and CO₂ gases were procured from BOC Gas 118 (Australia). The Reverse Osmosis (4 stages, IBIS Technology, 119 Australia) and Milli-Q Reference Water Purification System were 120 used to produce deionized water (18.2 M $\Omega \cdot$ cm at 25 °C).

The carbonization process involved taking approximately 2 grams of 122 the polymeric precursor and placing it in a ceramic crucible. The 123 crucible containing the sample was placed into a reactor and then 124 heated to a temperature of 803 K for a duration of 2 hours. Nitrogen gas 125 was employed as a carrier gas at a flow rate of around 2.0 dm³/min 126 (Figure 1(a)). The result of this carbonization process was 127 approximately 82% weight-loss of the precursor, which was determined 128 by measuring the sample's weight before and after carbonization. 129 Following carbonization, approximately 1 g of the resulting char was 130 subjected to the activation process. The activation method and specific 131 parameters used can be found in Table 1 and are schematically shown in 132 Figure 1.

An ultrasonic cone-shaped spray nozzle operating at 120 kHz (Sono-133 134 Tek Corp., USA) and set to a constant power of 3.0 W was employed 135 for the generation of micrometer-sized water droplets (Figure 1(b)). 136 Deionized water was delivered at a constant flow rate of 0.2 cm³/min 137 through a Teflon tube measuring 0.45 m in length with an inner 138 diameter 0.96 mm, facilitated by a syringe pump. It is important to note 139 that the activation process with highly concentrated supercritical steam 140 was accomplished by continually injecting water droplets generated by 141 the USN directly into the quartz tube reactor without the need for a 142 carrier gas flow (see movies in the Supporting Information). The 143 concentration of supercritical steam was gradually increased in a 144 controlled manner over 3 and 5 h, as illustrated in the bottom panel of 145 Figure 1(c). After 5 h of activation at 1173 K, the estimated density of 146 supercritical steam within the quartz tube reactor is approximately 147 0.031 g/cm³.

Low-Temperature Nitrogen Adsorption and Scanning 148 149 Electron Microscopy. The porosity of both the char and HPSCBs 150 was evaluated by using nitrogen adsorption-desorption isotherms at 77 151 K (Tristar II 3020, Micromeritics, USA). Prior to the measurements, 152 the samples were subject to vacuum degassing at 473 K for 24 h. The 153 pore size distributions (PSDs) and textural characteristics were assessed 154 using the quenched solid density functional theory (QSDFT) 155 method.²⁸⁻³⁰ A mixed slit and cylindrical kernel was used to model 156 equilibrium nitrogen adsorption in micro- and mesopores, respec-157 tively.²⁸⁻³⁰ The apparent Brunauer-Emmett-Teller (BET) surface 158 areas were determined by analyzing the nitrogen adsorption isotherms 159 within a relative pressure range of 0.05-0.3 (Table S1). Scanning 160 electron microscopy (SEM) observations and the energy-dispersive X-161 ray (EDX) analyses of carbon samples were conducted using a Verios 162 XHR SEM system at the Centre for Microscopy, Characterization, and 163 Analysis (CMCA) at the UWA. A slow-curing transparent epoxy known 164 as EpoFix was prepared by combining the resin and hardener in a 165 weight ratio of 25:3, respectively. Subsequently, carbon beads were 166 introduced into a 25 mm diameter mounting cup named FixiForm 167 along with the epoxy. The combined mixture was subjected to a vacuum 168 within a desiccator for a minimum duration of 24 h. Following this, the 169 epoxy molds containing the carbon beads underwent a polishing 170 process to reveal the interior of the beads. Subsequently, they were 171 coated with carbon before being imaged by SEM.

Small-Angle and Wide-Angle X-ray Scattering Measure- 172 ments. Combined small-angle X-ray scattering (SAXS) and wide-angle 173 X-ray scattering (WAXS) patterns were collected using a Nano-inXider 174 instrument from Xenocs SAS (Grenoble, France) with Cu K α radiation 175 and a two-detector setup for simultaneous measurements of the SAXS 176 and WAXS. Both detectors are Pilatus 300 k pixel detectors from 177 Dectris (Baden, Switzerland). The scattering intensity is a function of 178 the wavevector magnitude $q = 4\pi \sin(\theta)/\lambda$, with a specific value of X-ray 179 wavelength $\lambda = 0.154$ nm being used in this study. The samples were 180 loaded into vacuum-tight containers and sealed with $5-7 \,\mu\text{m}$ thick mica 181 windows. Each measurement was conducted under a vacuum, with an 182 acquisition time of 60 min. Standard reduction software (XSACT) was 183 used to average and correct the two-dimensional scattering data to yield 184 the data curves. The 1D SAXS and WAXS data were merged into a 185 single scattering curve, and the background from the mica windows was 186 subtracted (Figure S1). The theoretical treatment of the SAXS data is 187 described in detail in Supporting Information, Section 1. The standard 188 X-ray scattering diffraction analysis was conducted using a Rigaku 189 SmartLab instrument equipped with a HyPix-3000 detector. The X-ray 190 source used was Cu K α with a wavelength of 0.154 nm. The analysis 191 covered the angular range $10-90^{\circ}(2\theta)$ with a step of $0.01^{\circ}(2\theta)$ and a 192 scanning speed of 4°/min. A silicon zero diffraction holder was used for 193 the measurements (Figure S2). The spacing between the graphite layers 194 was determined using the Bragg equation³¹ (Table S1). 195

PFG NMR Diffusion Measurements. For NMR diffusion analysis, 196 the samples were prepared by immersing ~1.5 g in excess *n*-hexane 197 (>98%, Sigma-Aldrich). These samples were sealed tightly in 5 mm 198 NMR tubes (Bruker BioSpin) and left to soak for a minimum of 48 h. 199 The ¹H pulsed-field gradient (PFG) NMR measurements were 200 conducted by using a Magritek Spinsolve Diffusion spectrometer 201 (Magritek, New Zealand) equipped with a cylindrical 1 T Halbach 202 magnet array, providing a ¹H NMR frequency of 43 MHz. The 203 measurements were carried out at the default spectrometer temperature 204 of 301 K. Before analysis, all samples were allowed to reach thermal 205 equilibrium with the magnet bore for at least 20 min.

The diffusion of both unrestricted and imbibed *n*-hexane was 207 analyzed using the pulsed gradient stimulated echo (PGSTE) sequence, 208 as depicted in Figure 2. The self-diffusion coefficient of unrestricted *n*- 209 f2

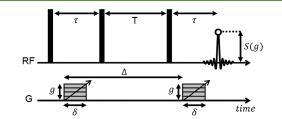


Figure 2. NMR pulse sequence diagram for the pulsed gradient stimulated echo (PGSTE) sequence used in this work. 90° radio frequency (RF axis) pulses are indicated by vertical bars. Gradient pulse timings (G axis) are specified according to the notation of Tanner;³³ magnetic field gradient pulses of incremental magnitude *g* are shown with pulse durations δ , the diffusion observation time is Δ , while *T* and τ represent the longitudinal storage period and spin echo time, respectively (Table S2).

hexane (D_0) was determined by fitting the acquired signal attenuation 210 data at low *b*-factor to³² 211

$$\frac{S(g)}{S(0)} = \exp(-bD_0)$$
(1) 212

where S(0) represents the NMR signal in the absence of any applied 213 field gradient, and S(g) is the acquired signal when the magnetic field 214 gradients of magnitude g are present (Figure S3). The *b*-factor is given 215 by 216

$$b = \gamma^2 g^2 \delta^2 \left(\Delta - \frac{\delta}{3} \right) \tag{2}_{217}$$

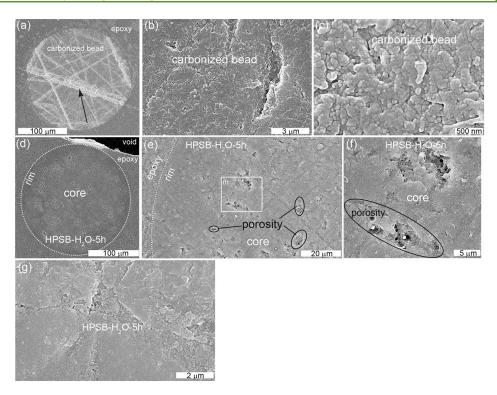


Figure 3. Secondary electron images of cross sections of char (a-c) and sample HPSCB-H₂O-Sh (d-g). (a) Low-magnification image of spherical char (~393 μ m in diameter). Note the numerous polishing scratches (black arrow) preserved in the bead, which document the relative softness of the material in comparison with the surrounding epoxy. (b-c) Higher-resolution image of the globular aggregates of the bead. This morphology is uniform across the entire sphere. (d) Poorly defined rim-core texture in sample HPSCB-H₂O-Sh. Note the relatively smooth external surface of the rim compared to the porous core. (e,f) Irregularly distributed areas/aggregates of macro- and mesopores of variable size (black ellipses) in the solid core (g). The dotted white line indicates the activated sphere, i.e., the contact between the sphere and epoxy.

218 where γ is the ¹H gyromagnetic ratio, while δ and Δ are the gradient 219 pulse duration and observation time, respectively, as shown in Figure 2. 220 A short Δ time of 20 ms was required to obtain an adequate NMR signal 221 in the presence of very short relaxation times, which arises from the 222 confinement of our *n*-hexane probe in carbon micropores.

223 The data collected from samples containing both char and carbon 224 beads immersed in excess *n*-hexane were fitted using a biexponential 225 equation of the following form^{33,34}

$$\frac{S(g)}{S(0)} = p_{inter} \exp(-bD_{inter}) + p_{intra} \exp(-bD_{intra})$$
(3)

227 where self-diffusion coefficients D_{inter} and D_{intra} describe signal 228 attenuation rates at small (interbead diffusion) and large (intrabead 229 diffusion) *b*, while p_{inter} and p_{intra} indicate the relative populations 230 characterized by these values. In each case, the measurements were 231 performed by holding the δ constant while linearly varying *g*.

²³² We calculated the PFG NMR interaction parameter, denoted as η_{intrav} ²³³ to get further insights into the intrabead *n*-hexane self-diffusion using ²³⁴ the following expression³⁴

$$\eta_{intra} = \frac{D_0}{D_{intra}}$$
(4)

236 where D_0 is the self-diffusion of *n*-hexane in the free bulk liquid, which is 237 determined to be 4.29·10⁻⁹ m²/s from experimental measurements (as 238 is outlined in Figure S3), and the intrabead self-diffusivity of *n*-hexane is 239 calculated from fitting eq 3 to the experimental data. The details of the 240 parameters used in the PGSTE measurements and the model used for 241 fitting eq 3 can be found in Table S2 and Section 2 of the Supporting 242 Information.

243 Kinetic Performance of Hierarchical Pore-Structured Carbon 244 Beads. Measurements of the Au recovery from cyanide solutions on 245 HPSCB samples were conducted to examine how the hierarchical pore 246 structures of carbon beads affect the kinetics of adsorption. Coconut

shell-based steam-activated carbon (Acticarb GC25PAU, Activated 247 Carbon Technologies Pty Ltd., Australia) was used as an industry 248 benchmark. The Acticarb sample used for comparison was selected to 249 have a size fraction in the range of approximately $351-500 \ \mu\text{m}$. This 250 choice was made to ensure that the granules of Acticarb would be 251 similar in size to those of the produced HPSCBs, facilitating a more 252 meaningful comparison. The carbon was initially ground by using a 253 mortar and pestle and then sieved to obtain the desired size fraction. A 254 gold standard solution with a concentration of 1 g/dm³ was obtained 255 from PerkinElmer (Glen Waverley, Vic, AUS). High-purity potassium 256 dicyanoaurate(I) (K[Au(Cn)₂]), with a purity of 98%, and ACS grade 257 Potassium Cyanide (KCN) were acquired from Sigma-Aldrich 258 (Truganina, Vic, AUS). The measurements using inductively coupled 259 plasma-optical emission spectroscopy (ICP-OES) were conducted 260 using a Thermo Scientific iCAP 7000 series instrument, specifically the 261 model 7600. The ICP-OES spectrometer was operated with the 262 following parameters: RF power, 1150 kW; auxiliary gas (Ar) flow, 0.5 263 dm³/min; nebulizer gas (Ar) flow, 0.5 dm³/min; exposure time, 15 s; 264 orientation, axial only; frequency, 500 Hz; replicates, 3. The 265 wavelengths selected for measuring Au metal ions were 242.795 and 266 267.595 nm. The 1 g/dm³ Au standard solution was diluted to 267 concetrations 10, 5, 2, and 1 mg/dm³, and these solutions were to create 268 the calibration curve. The calibration curve yielded a determination 269 coefficient of 0.999 for both Au wavelengths. The solution matrix used 270 for both the standards and the sample solutions was 0.2% HCl. A stock 271 feed solution of 500.0 mg/dm³ Au was prepared using 0.7315 g of 272 $K[Au(Cn)_2]$ with 12.5 g of KCN and NaOH. The pH of the stock feed 273 was adjusted to 2. This stock feed was used to prepare a 500 cm³ 274 solution with a concentration of 10.0 mg/dm^3 for each carbon sample. 275 An aggregate of 2 cm^3 was taken from each solution before inserting the 276 carbons and used as a 0 min reference. Approximately 0.125 g of carbon 277 materials wasplaced in a beaker with a magnetic stirrer and maintained 278 at 24 °C. At various time intervals (e.g., 10, 20, 30, 40, 50, 60, 80, 100, 279 120, 240, 480, 1440, and 2880 min), a 10 cm³ sample was collected from 280

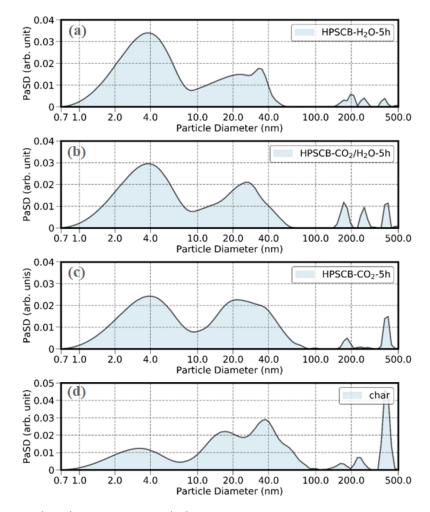


Figure 4. Particle size distributions (PaSD) for HPSCB samples (a-c) activated at 1173 K for 5 h using various activation methods (as outlined in Table 1) and char (d), calculated using SAXS data.

281 each solution for analysis. Each experimental data point represents
282 averages of three measurements. The average relative standard
283 deviation (RSD) error for the experimental data was less than 1.9%.
284 The kinetic data acquired from carbon samples were fitted to a
285 pseudo-first-order kinetic model with the following form³⁵

$$a(t) = a_e[1 - \exp(-k \cdot t)]$$
(5)

287 where *t* is the adsorption time, a_e and a(t) are the adsorption capacity of 288 $Au(CN)_2^-$ on the carbon sample at equilibrium and at time *t*, 289 respectively, and *k* is the pseudo-first-order rate constant.

290 RESULTS AND DISCUSSION

Morphological Changes. The discussion focuses on 291 292 morphological changes in HPSCB samples activated at 1173 K for 5 h. HPSCBs activated at 1173 K for 3 h exhibit a 293 294 morphology that closely resembles that of char. Figure 3 displays secondary (SE) and backscattered electron (BSE) images of 295 char and HPSCB-H₂O-5h. SEM images collected for HPSCB-296 CO₂-5h, HPSCB-CO₂/H₂O-5h, and corresponding energy-297 298 dispersive X-ray (EDS) spectra are shown in Section 3, 299 Supporting Information (Figures S4–S7). The activation 300 method does not affect the relative concentrations of S, O, 301 and C (e.g., S is $\sim 2.3 \pm 0.3$ wt %, the content of O is $\sim 2.1 \pm 0.7$ 302 wt %, and C is a dominant element with a relative concentration 303 of ~95.5 \pm 0.6 wt %). Notably, microscopic observations 304 indicate that each of the carbon beads produced exhibited a rim

approximately 10 to 25 μ m thick, encircling the core. No rim was 305 detected in the char. Although all carbon beads preserved their 306 external shape, the boundary/interface between the rim and core 307 exhibited variations. HPSCB-CO₂-Sh and HPSCB-CO₂/H₂O- 308 Sh both exhibited a distinct interface between the rim and the 309 core, whereas the HPSCB-H₂O-Sh displayed larger embayment 310 areas. Without additional characterization, it is not possible to 311 definitively attribute these embayments to the activation 312 process. 313

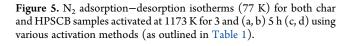
An in-depth analysis of the structural characteristics of the rim 314 and the associated parameters is beyond the scope of this paper. 315 However, it seems imperative to conduct additional research to 316 determine whether there are any differences between the rim 317 and the core. Discrepancies in characteristics such as porosity, 318 density, hardness, or permeability between these two regions 319 hold intriguing potential for modulating adsorption and 320 desorption kinetics. 321

Figures 4 and S8 illustrate the particle size distribution 322 f4 (PaSD) derived from the SAXS data. These distributions pertain 323 to HPSCB samples activated at 1173 K for 5 (a-c) and 3 h using 324 various activation methods, as well as char (d). Based on the 325 count and locations of the peaks, we can conclude that both char 326 and the produced HPSCB exhibit a distinctive three-level 327 hierarchical structure.^{36–40} The initial prominent symmetrical 328 peak at ~1–10 nm is ascribed to graphene nanoclusters.^{36–40} 329 Subsequent WAXS analysis verified the existence of turbostratic 330

331 graphene clusters, characterized by an average interlayer 332 distance of $d_{002} = 0.411 - 0.476$ nm (Table S1 and Figure S2). 333 The second major peak in the range of about 10-50 nm arises 334 from coherent scattering by nanoparticles.^{36–40} The $\sim 10-40$ 335 nm peak for HPSCB-H₂O-5h is broadened rather than a 336 symmetrical bell-shaped one (Figure 4). This suggests that the 337 excessive burnoff of char by highly concentrated supercritical 338 steam may lead to a more uniform distribution of nanoparticle 339 sizes. The third level of particulate hierarchy pertains to 340 aggregates of nanoparticles, with sizes that can be accessed via $_{341}$ SAXS, reaching up to 500 nm. $^{36-40}$ On this level, three peaks 342 were discerned, with char exhibiting the highest intensity and 343 HPSCB-H₂O-5h displaying the lowest. This suggests that the activation method has a relatively minor effect on the structure 344 of nanoparticle aggregates. 345

Accessible Pore Structure Changes. Figure 5 shows the nitrogen adsorption—desorption isotherms measured at 77 K for

f5



³⁴⁸ char and HPSCBs activated at 1173 K for 3 and (a, b) 5 h (c, d). ³⁴⁹ The N₂ adsorption–desorption isotherms confirm that all ³⁵⁰ samples have a hierarchical structure of a microporous matrix ³⁵¹ with embedded mesopore channels. Adsorption in micropores ³⁵² (typically smaller 1 nm) at low pressures (p/p₀ < 0.1) is ³⁵³ characterized by a sharp concave isotherm of type 1(a),⁴¹ while ³⁵⁴ adsorption in mesopores is characterized by a well-pronounced

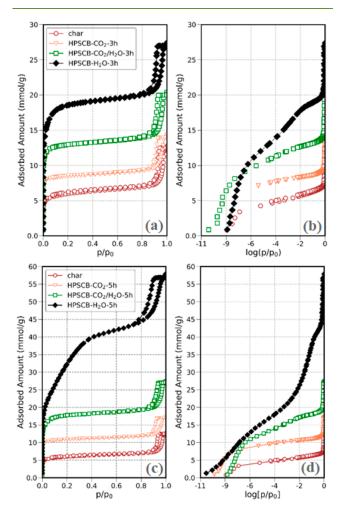
hysteresis loop at p/p0 > 0.8. The initial part of the isotherm on 355 the HPSCB-H₂O-3h sample is of type 1(b) that corresponds to 356 carbons with larger micropores and possible narrow mesopores 357 (<2.5 nm). Except for char, the H1-type adsorption-desorption 358 hysteresis loops of similar shape are closed at higher relative 359 pressures with no evidence of cavitation.^{41,42} The nitrogen 360 adsorption-desorption isotherms' reproducibility was validated 361 through independent measurements conducted on three 362 separate carbon samples (Figures S9-S11). The adsorption 363 and desorption isotherm for char exhibit partial irreversibility at 364 low pressures, suggesting kinetic limitations in the transport of 365 nitrogen molecules from the exterior to the pores. Additionally, a 366 weak swelling of the char filled with nitrogen is feasible. Detailed 367 examination of the PSD and the contributions of pore volume/ 368 surface area from micro- and mesopores (Figures 6-7 and Table 369 f6f7t2 2) highlights the substantial benefits of USN-assisted activation 370 t2 using highly concentrated supercritical steam. These benefits are 371 particularly evident in terms of the increased accessible surface 372 area and porosity. 373

The HPSCP-CO₂-3h sample, produced using traditional 374 activation with pure CO₂, is characterized by a sharp PSD with 375 an ultramicropore (<0.7 nm in width) fraction accounting for 376 93% of the accessible surface area, as indicated in Table 2. The 377 mesopore volume fraction (ranging from 2 to 50 nm in width), 378 at 41 vol % suggests that certain mesopores in the char have been 379 opened by CO2. The introduction of micrometer-sized water 380 droplets to the CO₂ carrier gas via USN injection leads to 381 significant changes in PSD, as depicted in Figure 6. The primary 382 peak in the PSD at 0.8 nm becomes broader as a result. The 383 majority of the accessible surface area, comprising 78%, can be 384 attributed to supermicropores (ranging from 0.7 to 2.0 nm in 385 width), while the volume fraction of accessible mesopores stands 386 at 38 vol % (Table 2). This increase in the fraction of 387 supermicropores can be explained by the merging of ultra- 388 micropore walls and the simultaneous formation of new 389 supermicropores from the early stages of the activation 390 process.43,44

The HPSCB-H₂O-3h sample produced through activation ³⁹² with USN-generated highly concentrated supercritical steam, ³⁹³ showcases the highest accessible pore volume ($0.88 \text{ cm}^3/\text{g}$) and ³⁹⁴ accessible surface area ($1272 \text{ m}^2/\text{g}$). These values are in line ³⁹⁵ with the highest char burnoff observed after 3 h of activation, as ³⁹⁶ detailed in Table 1. The PSD exhibits a broad peak at 0.9 nm, ³⁹⁷ accompanied by a distinct shoulder at 1.4 nm. Interestingly, the ³⁹⁸ fractions of accessible surface area attributed to ultra-, super-, ³⁹⁹ and mesopores closely resemble those of HPSCB-CO₂/H₂O-3h ⁴⁰⁰ (Table 2).

As we will demonstrate later, this outcome can be attributed 402 to the relatively short duration of the activation process. 403 Interestingly, extending the time of CO₂ activation from 3 to 5 h 404 does not yield significant changes in the PSD or the 405 corresponding fractions of accessible surface area and accessible 406 pore volume (Figures 6–7 and Table 2). Hence, our conclusion 407 is that the traditional 3 and 5 h of CO₂ activation of char at 1173 408 K is not sufficient for the alteration of the pore structure. 409 Likewise, there is no significant alteration in the pore structure 410 when extending the activation time from 3 to 5 h for carbon 411 beads produced using CO₂ enriched with USN-generated 412 micron-sized water droplets.

In contrast, subjecting char to 5 h of activation with USN- 414 generated highly concentrated supercritical steam at 1173 K 415 results in a substantial change in the nitrogen adsorption- 416 desorption isotherm, PSD, and accessible surface/volume ratio 417



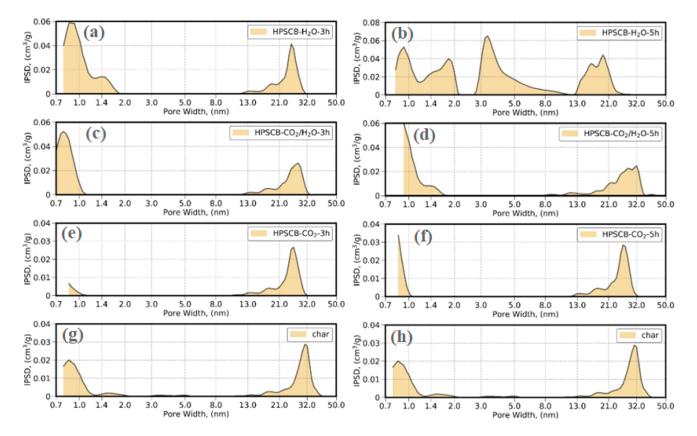


Figure 6. QSDFT pore size distributions for both char and HPSCBs activated at 1173 K for 3 (a, c, e, g) and 5 h (b, d, f, h) using various activation methods (as outlined in Table 1).

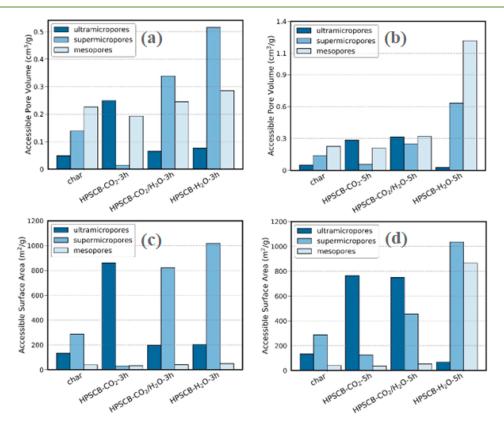


Figure 7. Alterations in the accessible QSDFT ultramicro-, supermicro-, and mesopore volume (a, b) and surface area (c, d) of both char and HPSCB samples activated at 1173 K for 3 (a, c) and 5 h (b, d) using various activation methods (as outlined in Table 1).

Table 2. Evolution of the Accessible Volume and Surface Area (*-access*), along with the Fractions of Relative Volume and Surface Area Pertaining to Accessible Ultramicropores (*-ultra*), Supermicropores (*-super*), and Mesopores (*-meso*) in Both Char and HPSCB Samples Activated at 1173 K for 3 and 5 h Using Various Activation Methods (as Outlined in Table 1)

Sample	V_{access} (cm ³ /g)	$rac{V_{ultra}}{V_{access}}$ (%)	$\frac{V_{super}}{V_{access}}$ (%)	$\frac{V_{meso}}{V_{access}}$ (%)	S_{access} (m ² /g)	$rac{S_{ultra}}{S_{access}}$ (%)	$\frac{S_{super}}{S_{access}}$ (%)	$\frac{S_{meso}}{S_{access}}$ (%)
char	0.41	12	34	56	460	29	62	9
HPSCB-CO ₂ -3h	0.46	54	2	41	923	93	3	4
HPSCB-CO ₂ /H ₂ O-3h	0.65	11	52	38	1058	19	78	4
HPSCB-H ₂ O-3h	0.88	9	59	33	1272	16	80	4
HPSCB-CO ₂ -5h	0.55	51	11	38	923	83	13	4
HPSCB-CO ₂ /H ₂ O-5h	0.88	35	28	36	1258	60	36	4
HPSCB-H ₂ O-5h	1.88	2	34	64	1967	3	53	44

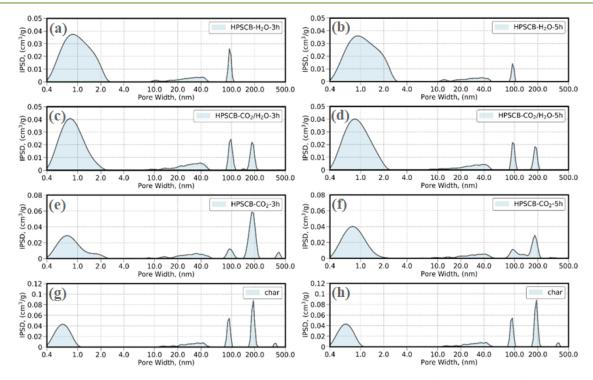


Figure 8. SAXS pore size distributions for both char and HPSCBs activated at 1173 K for 3 (a, c, e, g) and 5 h (b, d, f, h) using various activation methods (as outlined in Table 1).

418 of ultra-, super-, and mesopores, as evidenced by Figures 6-7 419 and Table 2. The linear shape of nitrogen adsorption isotherm 420 for HPSCB-H₂O-5h at very low pressures ($\sim 10^{-5} - 10^{-3} \text{ p/p}_0$) 421 serves as an indicator of the structural disorder present in the 422 pore walls (Figure 5).⁴⁵ The presence of linear segments on the 423 nitrogen adsorption isotherm suggests the organization of pores 424 into clusters. Indeed, the PSD calculated for HPSCB-H₂O-5h 425 reveals the presence of four broad and asymmetric peaks, 426 signifying pore clusters centered at 0.9, 1.8, 3.5, and 18 nm (as 427 shown in Figure 6). The substantial accessible pore volume $_{428}$ (1.88 cm³/g) and accessible surface area (1967 m²/g) of 429 HPSCB-H₂O-5h were finely tuned. The majority of the 430 accessible surface area, at 53%, is attributed to supermicropores, 431 followed by a 44% contribution from mesopores, while the contribution from ultramicropores is negligible (Table 2). In 432 terms of accessible volume, mesopores dominate with a 64 vol % 433 contribution, followed by supermicropores at 34 vol %, and a 434 435 minor 2 vol % contribution from ultramicropores, as shown in 436 Table 2.

437 **Open and Closed Surface Area Changes.** The pore size 438 distributions, extracted from SAXS measurements for char and HPSCBs activated at 1173 K for 3 and 5 h using various 439 activation methods (as summarized in Table 1), are presented in 440 Figure 8. In all carbon samples produced, SAXS PSD features a 441 f8 broad peak associated with micropores (~0.4-3.0 nm) and an 442 asymmetric peak corresponding to mesopores (\sim 20–40 nm), in 443 agreement with the QSDFT analysis (Figure 6). Furthermore, 444 there are discernible additional contributions from large 445 macropores, ranging from ~100-200 nm in width. Never- 446 theless, a careful inspection of SAXS PSDs reveals a diminishing 447 relative pore volume associated with these $\sim 100-200$ nm 448 macropores as char burnoff progresses. The enlargement and 449 coalescence of these macropores, likely facilitated by enhanced 450 gasification, may account for these observed patterns. It is worth 451 noting that the relative contributions of open (N₂-accessible) 452 and closed (SAXS data) surface areas are contingent upon both 453 time and the chosen activation method, as detailed in Table 3. 454 t3

The inaccessible surface area of char is notably large, reaching $_{455}$ approximately 84%. This high level of inaccessibility is due to the $_{456}$ fact that a significant portion of pores is opened during the $_{457}$ activation process. 43,44 Traditional activation with CO₂ at 1173 $_{458}$ K for 3 to 5 h only manages to open ~39–41% of the total $_{459}$

Table 3. Development of Open and Closed Porosity in HPSCBs Activated for 3 and 5 h Using Various Activation Methods (as Outlined in Table 1)^a

Sample	$S_{total} \left(m^2/g \right)$	S_{access} (m ² /g)	S _{closed} (%)	S _{open} (%)
char	2843	460	84	16
HPSCB-CO ₂ -3h	2339	923	61	39
HPSCB-CO ₂ /H ₂ O-3h	1841	1058	42	58
HPSCB-H ₂ O-3h	1568	1272	19	81
HPSCB-CO ₂ -5h	2233	923	59	41
HPSCB-CO ₂ /H ₂ O-5h	1483	1258	15	85
HPSCB-H ₂ O-5h	2138	1967	8	92

^{*a*}The values represented include the following: S_{total} - total surface area obtained from SAXS, S_{access} - accessible surface area obtained from QSDFT, and the fractions of closed (S_{closed}) and (S_{open}) surface areas obtained from combined SAXS-QSDFT analysis.

460 surface area, underscoring the inefficiency of this activation 461 method. The introduction of ultrasonic spraying of micrometer-462 sized water droplets to CO₂ gas promotes the opening of the $_{463}$ closed surface area in char. Following 5 h of activation with CO_2 464 gas enriched with USN-generated water droplets at 1173 K, 465 approximately 85% of the total surface area becomes accessible. 466 The most impressive results are achieved with USN-generated 467 highly concerted supercritical steam. After 5 h of activation and 468 enhanced gasification of char at 1173 K using this method, the 469 contribution from the accessible surface area increased 470 significantly to 92% (as outlined in Table 3). Notably, this 471 activation process using USN-generated highly concentrated 472 supercritical steam does not require the addition of a chemical 473 activation agent or a carrier gas flow. Moreover, the water 474 consumption is minimal, totaling 12 cm³ per 60 min of 475 activation.

Intrabead Transport Properties Changes. Figure 9 476 477 presents the experimental PFG NMR signal attenuation curves 478 of *n*-hexane-saturated char and HPSCBs activated at 1173 K for 479 3 (upper panels) and 5 h (bottom panels). Since diffusion rates 480 appear to be sufficiently separated (as shown in the left panels of 481 Figure 9), we utilized two independent straight lines to describe 482 the diffusion data (as expressed in eq 3).^{46,47} The initial rapid 483 decay observed is attributed to the unrestricted diffusion of *n*-484 hexane, which moves quickly between carbon beads (referred to 485 as interbead diffusion, Table 4).⁴⁶ The subsequent portion of the 486 attenuation curve reflects the diffusion of *n*-hexane confined ⁴⁸⁷ strictly within the pores (intrabead diffusion, Table 4).⁴⁶ It is 488 noteworthy that the uncertainty values for the PFG NMR 489 interaction parameter, obtained from eq 4, exhibit a range from approximately $\sim \pm 0.2\%$ to 1.1%. 490

The reduction in *n*-hexane self-diffusivity within the pores is 491 492 attributed to the limited displacement achievable within the 493 confined carbon nanospaces, as opposed to the more 494 unrestricted movement in bulk liquid ($\eta_{intra} > 1$, as shown in 495 Figure 9, right panel). The reduced diffusivity of *n*-hexane can be 496 attributed to several factors, including the tortuosity of the pore 497 network, adsorption interactions occurring at the pore surface, and activated diffusion through pore constrictions.⁴⁸ Upon 499 comparing PFG NMR signal attenuation curves for *n*-hexane in soo saturated char and HPSCB-CO₂-3h, it becomes evident that the 501 intrabead self-diffusion of *n*-hexane in these samples is highly so2 similar, as indicated by the η_{intra} values of 9.5–9.7 (Table 4). 503 This observation can be rationalized by the notion that char and 504 HPSCB-CO₂-3h share similar pore structures and pore 505 accessibility characteristics, as detailed in Tables 2 and 3.

Specifically, a short 3-h activation of char with CO2 at 1173 K 506 does not lead to substantial alterations in the pore network of the 507 char. Ultrasonic spraying of water droplets onto CO₂ gas or 508 directly into the furnace serves to enhance gasification. As a 509 consequence, there is a substantial increase in the accessible 510 porosity, as indicated in Table 2. Hence, self-diffusion of n- 511 hexane within the pores of HPSCB-CO₂/ H_2O -3h and HPSCB- 512 H_2O-3h is notably hindered (Figure 9 upper panels). In the case 513 of HPSCB-CO₂/H₂O-3h, the intrabead self-diffusion parameter 514for *n*-hexane is notably high at 21.7, suggesting severe 515 restrictions on the self-diffusion of *n*-hexane molecules within 516 the pores. However, when USN-assisted activation with highly 517 concentrated supercritical steam is utilized, which encourages 518 the development of open porosity with a substantial fraction of 519 accessible surface area and pore volume in the mesopores, the 520 intraparticle intrabead self-diffusion parameter for *n*-hexane is 521 reduced to 14.5. This reduction indicates an enhanced 522 connectivity of the hierarchical pore network within HPSCB- 523 H_2O-3h compared to HPSCB-CO₂/ H_2O-3h . 524

The extended 5-h activation of char with CO₂ at 1173 K 525 primarily results in the formation of ultramicropores (Table 2 526 and Figures 6-7). As expected, the intrabead self-diffusion of n- 527 hexane within this sample is considerably impeded, with an 528 interaction parameter of η_{intra} of 23.8. The introduction of water 529 droplets into CO2 gas via ultrasonic spraying over a 5-h period 530 enhances the self-diffusion of *n*-hexane within carbon pores, 531 resulting in a reduction of the interaction parameter η_{intra} to 16.7. 532 The highest intrabead diffusivity of *n*-hexane ($\eta_{intra} = 9.0$) was 533 achieved in the HPSCB-H₂O-5h sample, which was produced 534 through a 5-h activation of char with highly concentrated 535 supercritical steam at 1173 K. This result aligns with our 536 expectations, considering the high relative volume fraction of 537 mesopores (64 vol %), moderate volume fraction super- 538 micropores (34 vol %), negligible relative fraction of ultra- 539 micropores (2 vol %), and a very small relative fraction of 540 inaccessible surface area (8%) in the HPSCB-H₂O-5h sample 541 (Table 2). Interestingly, the slopes of the linear segments of our 542 PFG NMR signal attenuation curves, which correspond to 543 intrabead self-diffusion of *n*-hexane, are remarkably similar for 544 both char and HPSCB-H₂O-5h samples (Figure 9, bottom 545 panels).

The observation that restricted self-diffusion of *n*-hexane in 547 both char and HPSCB-H₂O-5h pores is quite similar (η_{intra} = 9.5 548 and 9.0, respectively, as shown in Figure 9) is noteworthy, 549 especially when considering the difference in accessible surface 550 area and accessible pore volume, which are approximately 1507 551 m²/g and 1.47 cm³/g, respectively (Tables 2 and 3). 552 Additionally, it is worth highlighting that the intrabead self- 553 diffusion coefficient of *n*-hexane in HPSCB-H₂O-5h (4.7 × 554 10⁻¹⁰ m²/s) is comparable to the intraparticle self-diffusion 555 coefficients of *n*-hexane in various porous materials, such as the 556 NaX zeolite crystal (6.4 × 10⁻¹⁰ m²/s),⁴⁹ mesoporous silica type 557 MCM-41 with regular 3.8 nm mesopores (4.1 × 10⁻¹⁰ m²/s),⁴⁶ 558 and MOF-5 crystal (4.0 × 10⁻¹⁰ m²/s).⁵⁰

Adsorption Kinetic Performance. The PFG NMR 560 intrabead *n*-hexane self-diffusion data offer valuable insights 561 into the mobility of individual molecules within the pore 562 networks under equilibrium conditions. Our findings have 563 shown that a high relative volume fraction of mesopores 564 enhances the self-diffusion of *n*-hexane within the pore networks 565 of hierarchically structured carbon beads (Figure 9). Con- 566 sequently, we anticipate that the hierarchical pore structure of 567 the produced carbon beads could potentially improve the 568

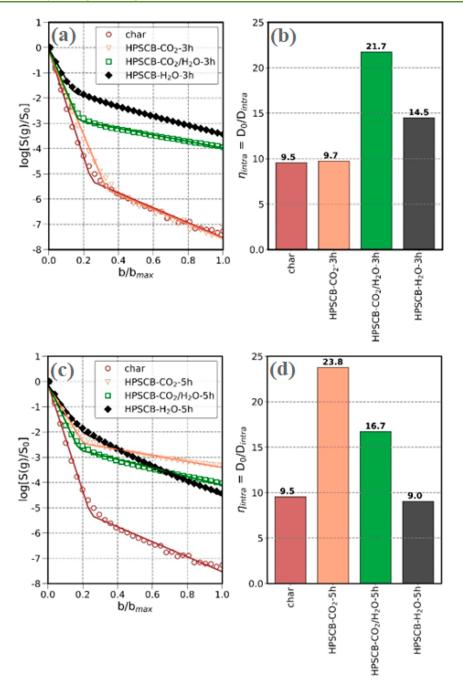


Figure 9. Left panels (a, c) show both the experimental (symbols) and theoretical (solid lines) PFG NMR signal attenuation curves of *n*-hexane saturated char and HPSCB samples activated at 1173 K for 3 (a, b) and 5 h (c, d) using various activation methods (as outlined Table 1). Right panels (b, d) display the PFG NMR interaction parameter for intrabead *n*-hexane self-diffusion in the porous networks of the carbon materials that were produced (Table 4).

569 kinetics of ion adsorption from aqueous solutions. In order to 570 evaluate our hypothesis, we conducted experiments to measure 571 the adsorption of gold from cyanide solutions onto different 572 carbon materials: char, HPSCBs activated at 1173 for 5 h, and 573 the industrial activated carbon Acticarb GC25PAU, which is 574 commonly employed in the carbon-in-pulp process.

⁵⁷⁵ HPSCB-H₂O-5h exhibits the fastest kinetics of gold ⁵⁷⁶ adsorption, as evidenced by the pseudo-first-order rate constant ⁵⁷⁷ of 4.48×10^{-3} 1/min (Figure 10 and Table 5). In contrast, the ⁵⁷⁸ gold adsorption kinetics in Acticarb GC25PAU is the slowest (k ⁵⁷⁹ = 2.89×10^{-3} 1/min, Figure 10, and Table 5). This is consistent with Acticarb GC25PAU having a negligible volume fraction of 580 accessible mesopores (0.5 vol %) and a negligible fraction of 581 accessible surface area of mesopores (0.7%) (Figure S12(a)). 582 Analyzing the experimental results reveals that HPSCB-H₂O-5h 583 achieves a 29% and 41% higher gold uptake compared to the 584 benchmark Acticarb GC25PAU activated carbon after 60 and 585 120 minutes, respectively (Figure 10). Both HPSCB-CO₂-5h 586 and HPSCB-CO₂/H₂O-5h adsorbed the gold cyanide complex 587 faster than did Acticarb GC25PAU activated carbon. This trend 588 aligns with the highly obstructed self-diffusion of *n*-hexane in 589 Acticarb GC25PAU micropores (η_{intra} = 40.8 obtained from 590 Table 4. Intrabead (D_{intra}) and Interbead (D_{inter}) Self-Diffusion Coefficients of *n*-Hexane in Char and HPSCBs Activated at 1173 K for 3 and 5 h Using Various Activation Methods (as Outlined Table 1) Obtained from PFG NMR at 301 K^{*a*}

Sample	$D_{inter} \left(\mathrm{m}^2 / \mathrm{s} \right)$	D_{intra} (m ² /s)	η_{intra}	R^2
char	3.21×10^{-9}	4.50×10^{-10}	9.54	0.9949
HPSCB-CO ₂ -3h	2.49×10^{-9}	4.41×10^{-10}	9.73	0.9942
HPSCB-CO ₂ /H ₂ O- 3h	2.49×10^{-9}	1.97×10^{-10}	21.73	0.9939
HPSCB-H ₂ O-3h	1.85×10^{-9}	2.96×10^{-10}	14.48	0.9964
HPSCB-CO ₂ -5h	1.71×10^{-9}	1.8×10^{-10}	23.76	0.9895
HPSCB-CO ₂ /H ₂ O- 5h	2.23×10^{-9}	2.57×10^{-10}	16.71	0.9932
HPSCB-H ₂ O-5h	1.43×10^{-9}	4.47×10^{-10}	9.05	0.9959
ATL DEC NIME		1	1 1	10

"The PFG NMR interaction parameter for intrabead *n*-hexane selfdiffusion is computed from eq 4, and R^2 denotes the coefficient of determination.

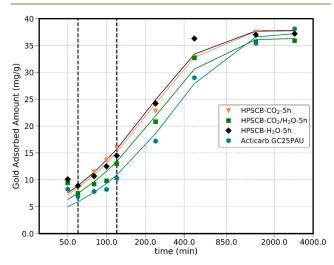


Figure 10. Adsorption kinetics of $Au(CN)_2^-$ on HPSCBs activated at 1173 K for 5 h using various activation methods (as outlined in Table 1) and Acticarb GC25PAU, a commercial activated carbon, measured at pH 2 and 297 K. HPSCB-H₂O-Sh showed 29% and 41% higher gold uptake than the benchmark Acticarb GC25PAU industrial activated carbon after 60 and 120 min, respectively (dashed lines).

Table 5. Pseudo-First-Order Kinetic Model Coefficients for the Adsorption (pH of 2 and 297 K) of $Au(CN)_2^-$ on HPSCB Samples Activated for 5 h at 1173 K Using Various Activation Methods (as Outlined in Table 1)^{*a*}

	Sample	$a_e (\mathrm{mg/g})$	$k(1/\min)$	R^2
	HPSCB-CO ₂ -5h	37.7	4.27×10^{-3}	0.996
	HPSCB-CO ₂ /H ₂ O-5h	36.3	3.86×10^{-3}	0.984
	HPSCB-H ₂ O-5h	37.8	4.48×10^{-3}	0.985
	Acticarb GC25PAU	37.2	2.89×10^{-3}	0.985
aт	The coefficients for Act	icarb GC25PA	U. a commercia	l activated

carbon, are also displayed for comparison.

591 PFG NMR measurements of *n*-hexane saturated Acticarb
592 GC25PAU, Figure S12(b)-(c) in the Supporting Information).
593 Considering the crucial role of gold cyanide complex
594 adsorption kinetics in the carbon-in-pulp process,²⁶ it is clear
595 that the kinetics of gold adsorption by HPSCBs is highly
596 promising. Columns packed with HPSCB-H₂O-Sh spherical
597 carbon beads show significant potential for enhancing the

efficiency of gold recovery. This indicates that HPSCB-H₂O-5h $_{598}$ has the potential to significantly enhance the gold extraction $_{599}$ process, making it a valuable asset in gold mining and processing $_{600}$ operations.

The Energy Cost of Steam Generation: USN versus the 602 Water Heating Process. Figure 11 illustrates the relationship 603 ful

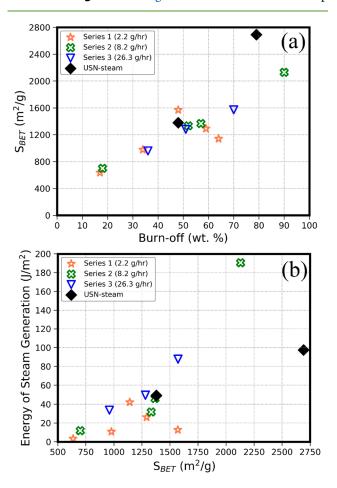


Figure 11. (a) Variation in the apparent BET surface area of activated carbons produced via physical activation with USN and water heating steam generation as a function of burnoff. (b) Energy consumption vs the apparent BET surface area for steam generation via the USN and water heating in the physical activation of char at 1173 K. Steam activation conditions:⁵¹ Set 1: water bath temperature of 333 K, activation temperature 1173 K, and 2.2 g/h steam rate; Set 2: water bath temperature of 353 K, activation temperature 1173 K, and 8.2 g/h steam rate; Set 3: water bath temperature of 363 K, activation temperature 1173 K, and 26.3 g/h steam rate.

between the apparent BET surface area of activated carbons 604 produced through the USN and traditional steam activation at 605 1173 K, as well as the burnoff degree. In our calculations, we 606 made use of published data pertaining to three series of activated 607 carbons. These activated carbons were produced with varying 608 steam flow rates that originated from the water heating process 609 (Section 4 in the Supporting Information).^{S1} To maintain 610 consistency when comparing the energy consumption for steam 611 generation estimated from both the current experimental data 612 and previously published results,⁵¹ it is crucial to highlight that 613 all calculations related to the energy cost of steam generation 614 have been standardized per square meter of the apparent BET 615 surface area. One of the primary conclusions drawn from our 616 analysis is that carbon beads, which undergo a 5-h activation 617 618 process using USN-generated highly concentrated supercritical 619 steam at 1173 K, exhibit the highest apparent BET surface area, 620 measuring 2691 m²/g. This level of surface area is achieved at a 621 burnoff degree of approximately 79 ± 11 wt % (Figure 11(a)). 622 The produced spherical carbon beads, averaging $238 \pm 1.4 \mu m$ 623 in diameter, exhibit mechanical stability (Table S1). The 624 activated carbon, resulting from 6 h of traditional steam 625 activation at 1173 K, achieves its highest apparent BET surface 626 area of 2130 m²/g at a burnoff of 90 wt % (Figure 11(a)). It is 627 important to note that this activated carbon is in powder form, 628 attributed to the disintegration of carbon beads, as reported by 629 the authors.⁵¹

The energy cost of USN and water heating steam generation 630 631 for producing high surface area activated carbons ($S_{\text{BET}} > \sim 2000$ $_{632}$ m²/g) at 1173 K varies significantly. The estimated energy 633 required for USN steam generation to develop the 1 m² apparent 634 BET surface area for HPSCB-H₂O-5h is 97 J. On the other hand, 635 the estimated energy required for water heating steam 636 generation to develop the 1 m² apparent BET surface area for 637 powdered activated carbon with $S_{\text{BET}} = 2130 \text{ m}^2/\text{g}$ is 190 J 638 (Figure 11(b)). For activated carbons with S_{BET} around 1000– $639 1600 \text{ m}^2/\text{g}$, the energy cost of both USN and water heating 640 steam generation is similar, approximately ranging from 30 to 50 $_{641}$ J/m². This is evident from the cluster of experimental points $_{642}$ shown in Figure 11(b). These findings consistently demonstrate 643 that the combustion kinetics and reaction pathways between the 644 carbon structure and steam at 1173 K are influenced by the 645 method of steam generation, particularly when the burnoff of 646 char is high, surpassing approximately 70 wt % as illustrated in 647 Figure 11.

648 CONCLUSIONS

649 We developed an ultrasonic spray nozzle-quartz tube reactor to 650 produce hierarchical pore-structured carbon beads (HPSCBs). 651 Our innovative approach offers several advantages, including 652 increased energy efficiency per square meter of the apparent 653 BET surface area, reduced activation time and water 654 consumption, and improved gold recovery kinetics when 655 compared to the industrial benchmark. In particular, the 656 production of high surface area HPSCBs ($S_{BET} > \sim 2000 \text{ m}^2/$ 657 g) using USN-generated highly concentrated supercritical steam 658 consumes approximately 97 J/m² for steam generation. In 659 contrast, the estimated energy cost of steam generation using 660 traditional steam activation is ~190 J/m².

661 Our study demonstrated that 5 h of activation with USN-662 generated highly concentrated supercritical steam at 1173 K 663 outperforms traditional steam activation. In a previous study,⁵¹ 664 which used traditional steam activation at the same temperature 665 (1173 K), the maximum apparent BET surface area achieved 666 was 2130 m²/g. However, this process resulted in the breakdown 667 of spherical carbon beads, producing an activated carbon 668 powder. In comparison, our innovative reactor design allowed us 669 to produce the HPSCB-H₂O-5h sample with an increased 670 apparent BET surface area of 2691 m²/g, requiring 1 h less of 671 activation time, and preserving the structural integrity of the 672 carbon beads, indicating superior mechanical stability.

⁶⁷³ Pulse field gradient NMR revealed that the HPSCB-H₂O-Sh ⁶⁷⁴ sample exhibits the fastest intrabead self-diffusivity of *n*-hexane ⁶⁷⁵ at 301 K (η_{intra} = 9.0) compared to all other carbon samples, ⁶⁷⁶ indicating that supermicro- and mesopore clusters are well-⁶⁷⁷ connected. The intrabead self-diffusion of *n*-hexane in HPSCB-⁶⁷⁸ H₂O-Sh (4.7 × 10⁻¹⁰ m²/s) is comparable with intraparticle self-⁶⁷⁹ diffusion of *n*-hexane in the NaX zeolite crystal (6.4×10^{-10} m²/ s),⁴⁹ mesoporous silica type MCM-41 with regular 3.8 nm $_{680}$ mesopores (4.1 \times 10⁻¹⁰ m²/s),⁴⁶ and MOF-5 crystal (4.0 \times $_{681}$ $10^{-10} \text{ m}^2/\text{s}$).⁵⁰ The exceptional intrabead self-diffusion rate of *n*- 682 hexane in the HPSCB-H2O-5h sample can be attributed to its 683 distinct hierarchical pore structure, which predominantly 684 comprises supermicropores (accessible surface area 53%, 685 accessible pore volume 34 vol %, Table 2) and mesopores 686 (accessible surface area 44%, accessible pore volume 64 vol %, 687 Table 2). The intrabead self-diffusion of *n*-hexane in char (4.5×688) $10^{-10} \text{ m}^2/\text{s}$) is slightly higher compared to HPSCB-H2O-5h, yet 689 it remains notably rapid. This experimental result can be 690 attributed to the observation that the hierarchical structure of 691 char consists of a low fraction of ultramicropores (with an 692 accessible surface area of 29% and an accessible pore volume of 693 12 vol %, as presented in Table 2). Furthermore, since 694 approximately 84% of the total surface area is closed, we 695 anticipate that all accessible pores are distributed around the 696 char's surface, which may enhance the self-diffusion of *n*-hexane. 697 To further investigate and confirm our hypothesis, additional 698 PFG-NMR research employing different probes is necessary. 699

The kinetic measurements of gold dicyanide recovery clearly 700 demonstrated that HPSCB-H₂O-5h, characterized by inter-701 connected supermicroporous and mesoporous regions, is a 702 superior carbon adsorbent. HPSCB-H₂O-5h showed a 29% and 703 41% higher gold uptake than the benchmark Acticarb 704 GC25PAU industrial activated carbon after 60 and 120 min. 705 This underscores the enhanced adsorption kinetics and 706 efficiency of HPSCB-H₂O-5h, making it a promising material 707 for gold recovery applications. 708

The implementation of USN-assisted steam activation has led 709 to a noteworthy decrease in energy demands, allowing for the 710 production of high-quality carbon beads in a shorter time. The 711 development of highly concentrated supercritical steam 712 activation presents an exciting prospect for the sustainable 713 fabrication of tailored activated carbon featuring adjustable 714 hierarchical pore architectures and remarkably large surface 715 areas. 716

ASSOCIATED CONTENT 717

718

Supporting Information

The Supporting Information is available free of charge at 719 https://pubs.acs.org/doi/10.1021/acssuschemeng.3c04171. 720

Small-angle X-ray spectrum (SAXS) data treatment for 721 analysis of pore and particle size distributions, SAXS 722 diffraction patterns, X-ray diffraction angle (XRD) 723 patterns, description of the procedures for collecting 724 and processing pulse-field gradient nuclear magnetic 725 resonance (PFG NMR) data, scanning electron micros- 726 copy (SEM) images, energy dispersive X-ray spectroscopy 727 (EDX) analysis, particle size distributions (PaSD) for 728 char and HPSCB samples activated for 3 h at 1173 K, N2 729 adsorption-desorption isotherms at 77 K, PFG NMR 730 experimental results for Acticarb GC25PAU activated 731 carbon, distributions of bead sizes obtained from optical 732 microscopy, energy calculation for ultrasonic nozzle 733 steam generation (PDF) 734 Video depicting the process of ultrasonic spraying of 735 water droplets into the quartz tube reactor (MP4) 736 Video depicting the process of ultrasonic spraying of 737 water droplets into the quartz tube reactor (MP4) 738

Video depicting the process of ultrasonic spraying of 739 water droplets into the quartz tube reactor (MP4) 740

pubs.acs.org/journal/ascecg

741 Video depicting the process of ultrasonic spraying of742 water droplets into the quartz tube reactor (MP4)

743 **AUTHOR INFORMATION**

744 Corresponding Author

- 745 Piotr Kowalczyk School of Mathematics, Statistics,
- 746 Chemistry, and Physics, Murdoch University, Murdoch,
- 747 Western Australia 6150, Australia; o orcid.org/0000-0002-
- 748 7523-0906; Phone: +61 8 9360 6936;
- 749 Email: P.Kowalczyk@murdoch.edu.au

750 Authors

- 751 Michael Hough School of Mathematics, Statistics, Chemistry,
- and Physics, Murdoch University, Murdoch, Western Australia
 6150, Australia
- Artur Deditius School of Mathematics, Statistics, Chemistry,
 and Physics, Murdoch University, Murdoch, Western Australia
- ⁷⁵⁶ 6150, Australia; School of Earth Sciences, The University of
- 757 Western Australia, Perth, Western Australia 6009, Australia
- 758 Neil Robinson Department of Chemical Engineering, The
- University of Western Australia, Perth, Western Australia
 6009, Australia; orcid.org/0000-0002-0893-2190
- Gerd Elmar Schröder-Turk School of Mathematics,
- 762 Statistics, Chemistry, and Physics, Murdoch University,
- 763 Murdoch, Western Australia 6150, Australia; Department of
- 764 Materials Physics, Australian National University, Canberra,
- 765 Australian Capital Territory 2601, Australia
- 766 Jacob Judas Kain Kirkensgaard Department of Food Science,
- 767 University of Copenhagen, 1958 Frederiksberg, Denmark;
 768 Niels Bohr Institute, University of Copenhagen, 2100
- 769 Copenhagen, Denmark; © orcid.org/0000-0001-6265-0314
- 770 Volodymyr M. Gun'ko Chuiko Institute of Surface
- 771 *Chemistry, 03164 Kyiv, Ukraine;* ^(a) orcid.org/0000-0001-772 6333-3441
- 773 Alexander V. Neimark Department of Chemical and
- 774 Biochemical Engineering, Rutgers, The State University of New
- 775 Jersey, Piscataway, New Jersey 08854-8058, United States;
 776 orcid.org/0000-0002-3443-0389
- 777 Katsumi Kaneko Research Initiative for Supra-Materials,
- 778 Shinshu University, Nagano 380-8553, Japan; [©] orcid.org/ 779 0000-002-2378-478X

780 Complete contact information is available at:

781 https://pubs.acs.org/10.1021/acssuschemeng.3c04171

782 Author Contributions

The manuscript was written through contributions of all 783 authors. All authors have given approval to the final version of 784 the manuscript. Michael Hough: Conceptualization, Inves-785 tigation, Methodology, Data curation, Software, Funding 786 acquisition. Artur Deditius: Methodology, Data curation. Neil 787 Robinson: Investigation, Methodology, Data curation. Gerd 788 Elmar Schröder-Turk: Conceptualization, Data curation. Jacob 789 790 Judas Kain Kirkensgaard: Investigation, Methodology, Data curation. Volodymyr M. Gun'ko: Methodology, Software, Data 791 curation. Alexander V. Neimark: Software, Data curation. 792 Katsumi Kaneko: Data curation. Piotr Kowalczyk: Project 793 794 administration, Conceptualization, Data curation.

795 Notes

796 The authors declare no competing financial interest.

ACKNOWLEDGMENTS

We would like to acknowledge the support from the Australian 798 Microscopy and Microanalysis research facility at the Centre for 799 Microscopy, Characterisation, and Analysis, University of 800 Western Australia. We would like to thank M. Kowalczyk for 801 assistance with ICP-OES measurements. We acknowledge 802 Nicholas Corrente and Shivam Parashar, Rutgers University, 803 for their help with PSD calculations. 804

REFERENCES

(1) Bandosz, T. J., Ed.; Activated Carbon Surfaces in Environmental 806 Remediation; Elsevier: Amsterdam, 2006. 807

(2) Pomerantseva, E.; Bonaccorso, F.; Feng, X.; Cui, Y.; Gogotsi, Y. 808 Energy storage: the future enabled by nanomaterials. *Science* **2019**, *366*, 809 6468. 810

(3) Nanaji, K.; Upadhyayula, V.; Rao, T. N.; Anandan, S. Robust, 811 Environmentally Benign Synthesis of Nanoporous Graphene Sheets 812 from Biowaste for Ultrafast Supercapacitor Application. *ACS Sustain.* 813 *Chem. Eng.* **2019**, 7 (2), 2516–2529. 814

(4) Ichikawa, T.; Matsuo, T.; Tachikawa, T.; Yamada, T.; Yoshimura, 815 T.; Yoshimura, M.; Takagi, Y.; Sawama, Y.; Sugiyama, J.-I.; Monguchi, 816 Y.; Sajiki, H. Microwave-Mediated Site-Selective Heating of Spherical-817 Carbon-Bead-Supported Platinum for the Continuous, Efficient 818 Catalytic Dehydrogenative Aromatization of Saturated Cyclic Hydro-819 carbons. ACS Sustain. Chem. Eng. **2019**, 7 (3), 3052–3061. 820

(5) Suresh, S.; Bandosz, T. J. Removal of formaldehyde on carbon- 821 based materials: A review of the recent approaches and findings. *Carbon* 822 **2018**, 137, 207–221. 823

(6) Zhang, En; Fulik, N.; Hao, G.-P.; Zhang, H.-Y.; Kaneko, K.; 824 Borchardt, L.; Brunner, E.; Kaskel, S. An Asymmetric Supercapacitor- 825 Diode (CAPode) for Unidirectional Energy Storage. *Angew. Chem., Int.* 826 *Ed.* **2019**, *58*, 13060–13065. 827

(7) Kowalczyk, P.; Holyst, R.; Terrones, M.; Terrones, H. Hydrogen 828 storage in nanoporous carbon materials: myth and facts. *Phys. Chem.* 829 *Chem. Phys.* **2007**, *9*, 1786–1792. 830

(8) Wei, L.; Tian, K.; Zhang, X.; Jin, Y.; Shi, T.; Guo, X. 3D Porous 831 Hierarchical Microspheres of Activated Carbon from Nature through 832 Nanotechnology for Electrochemical Double-Layer Capacitors. *ACS* 833 *Sustain. Chem. Eng.* **2016**, *4* (12), 6463–6472. 834

(9) Goldfarb, J. L.; Dou, G.; Salari, M.; Grinstaff, M. W. Biomass- 835
Based Fuels and Activated Carbon Electrode Materials: An Integrated 836
Approach to Green Energy Systems. ACS Sustain. Chem. Eng. 2017, 5 837
(4), 3046–3054. 838

(10) Kowalczyk, P.; Terzyk, A. P.; Erwardt, P.; Hough, M.; Deditius, 839 A. P.; Gauden, P. A.; Neimark, A. V.; Kaneko, K. Machine learning- 840 assisted design of porous carbons for removing paracetamol from 841 aqueous solutions. *Carbon* **2022**, *198*, 371–381. 842

(11) Marsh, H.; Rodríguez-Reinoso, F. *Activated Carbon*; Elsevier: 843 Amsterdam, 2006. 844

(12) Inagaki, M. New Carbons - Control of Structure and Functions; 845 Elsevier Science Ltd.: Amsterdam, 2000. 846

(13) Leng, J.; Wang, Z.; Wang, J.; Wu, H.-H.; Yan, G.; Li, X.; Guo, H.; 847 Liu, Y.; Zhang, Q.; Guo, Z. Advances in nanostructures fabricated via 848 spray pyrolysis and their applications in energy storage and conversion. 849 *Chem. Rev. Soc.* **2019**, *48*, 3015–3072. 850

(14) Shiflett, M. B.; Foley, H. C. Ultrasonic deposition of high- 851 selectivity nanoporous carbon membranes. *Science* **1999**, 285, 1902– 852 1904. 853

(15) Jeong, C.; Ko, H.; Kim, H.-T.; Sun, K.; Kwon, T.-H.; Jeong, H. E.; 854 Park, Y.-B. Bioinspired, high-sensitivity mechanical sensors realized 855 with hexagonal microcolumnar arrays coated with ultrasonic-sprayed 856 single-walled carbon nanotubes. *ACS Appl. Mater. Interfaces* **2020**, *12*, 857 18813–18822. 858

(16) Song, Y. L.; Tsai, S. C.; Chen, C. Y.; Tseng, T. K.; Tsai, C. S.; 859 Chen, J. W.; Yao, Y. D. Ultrasonic spray pyrolysis for synthesis of 860 spherical zirconia particles. *J. Am. Chem. Soc.* **2004**, 87, 1864–1871. 861 (17) Skrabalak, S. E.; Suslick, K. S. Porous MoS2 synthesized by 862 ultrasonic spray pyrolysis. *J. Am. Chem. Soc.* **2005**, *127*, 9990–9991. 863

805

797

864 (18) Bang, J. H.; Suslick, K. S. Applications of ultrasound to the 865 synthesis of nanostructured materials. *Adv. Mater.* **2010**, *22*, 1039– 866 1059.

(19) Gadipelli, S.; Howard, C. A.; Guo, J.; Skipper, N. T.; Zhang, H.;
Shearing, P. R.; Brett, D. J. L. Superior multifunctional activity of
nanoporous carbons with widely tunable porosity: enhanced storage
capacities for carbon-dioxide, hydrogen, water, and electric charge. *Adv. Energy Mater.* 2020, *10*, 1903649.

(20) Bianco, A.; Chen, Y.; Frackowiak, E.; Holzinger, M.; Koratkar,
N.; Meunier, V.; Mikhailovsky, S.; Strano, M.; Tascon, J. M. D.;
Terrones, M. Carbon science perspective in 2020: Current research and
future challenges. *Carbon* 2020, *161*, 373–391.

(21) Mikhalovsky, S. V.; Sandeman, S. R.; Howell, C. A.; Phillips, G. J.;
Nikolaev, V. G. Biomedical applications of carbon adsorbents. In *Novel Carbon Adsorbents*; Tascón, J. M. D., Ed.; Elsevier Ltd.: Oxford, 2012;
pp 639–669.

880 (22) Malik, D. J.; Warwick, G. L.; Mathieson, I.; Hoenich, N. A.; 881 Streat, M. Structured carbon haemoadsorbents for the removal of 882 middle molecular weight toxins. *Carbon* **2005**, *43*, 2317–2329.

(23) Davankov, V.; Pavlova, L.; Tsyurupa, M.; Brady, J.; Balsamo, M.;
Yousha, E. Polymeric adsorbent for removing toxic proteins from blood
of patients with kidney failure. *J. Chromatogr. B* 2000, 739, 73–80.

(24) Melillo, M.; Gun'ko, V. M.; Tennison, S. R.; Mikhalovska, L. I.;

7 Phillips, G. J.; Davies, J. G.; Lloyd, A. W.; Kozynchenko, O. P.; Malik, D.
887 J.; Streat, M.; Mikhalovsky, S. V. Structural characteristics of activated
889 carbons and ibuprofen adsorption affected by bovine serum albumin.
890 Langmuir 2004, 20, 2837–2851.

891 (25) Mikhalovsky, S. V. Emerging technologies in extracorporeal 892 treatment: focus on adsorption. *Perfusion* **2003**, *18*, 47–54.

893 (26) Nicol, M. J.; Fleming, C. A.; Cromberge, G. The absorption of 894 gold cyanide onto activated carbon. I. The kinetics of absorption from 895 pulps. J. S. Afr. Inst. Min. Metal. **1984**, 84, 50–54.

(27) Kowalczyk, P.; Deditius, A.; Ela, W. P.; Wiśniewski, M.; Gauden,
P. A.; Terzyk, A. P.; Furmaniak, S.; Wloch, J.; Kaneko, K.; Neimark, A.
V. Super-sieving effect in phenol adsorption from aqueous solutions on
nanoporous carbon beads. *Carbon* 2018, 135, 12–20.

900 (28) Neimark, A. V.; Lin, Y.; Ravikovitch, P. I.; Thommes, M. 901 Quenched solid density functional theory and pore size analysis of 902 micro-mesoporous carbons. *Carbon* **2009**, *47*, 1617–1628.

903 (29) Gor, G. Y.; Thommes, M.; Cychosz, K. A.; Neimark, A. V. 904 Quenched solid density functional theory method for characterization 905 of mesoporous carbons by nitrogen adsorption. *Carbon* **2012**, *50*, 906 1583–1590.

907 (30) Landers, J.; Gor, GYu; Neimark, A. V. Density functional theory 908 methods for pore structure characterization: Review. *Colloids Surf., A* 909 **2013**, *437*, 3–32.

910 (31) Warren, B. E. X-ray diffraction in random layer lattices. *Phys. Rev.*911 1941, 59, 693–698.

(32) Stejskal, E. O.; Tanner, J. E. Spin diffusion measurements: Spin
echoes in the presence of a time-dependent field gradient. *J. Chem. Phys.*1965, 42, 288–292.

915 (33) Tanner, J. E. Use of the Stimulated Echo in NMR Diffusion 916 Studies. J. Chem. Phys. **1970**, 52, 2523–2526.

917 (34) Price, W. S. NMR studies of translational motion: principles and 918 applications; Cambridge University Press: Cambridge, 2009.

919 (35) Wang, H.; Yuan, X.; Wu, Y.; Huang, H.; Zeng, G.; Liu, Y.; Wang, 920 X.; Lin, N.; Qi, Y. Adsorption characteristics and behaviors of graphene 921 oxide for Zn(II) removal from aqueous solution. *Appl. Surface Sci.* **2013**, 922 279, 432–440.

923 (36) Bansal, R. C.; Donnet, J. B.; Stoeckli, F. *Active Carbon*; Marcel 924 Dekker: New York, 1988.

925 (37) Gun'ko, V. M.; Meikle, S. T.; Kozynchenko, O. P.; Tennison, S. 926 R.; Ehrburger-Dolle, F.; Morfin, I.; Mikhalovsky, S. V. Comparative 927 characterization of carbon and polymer adsorbents by SAXS and 928 nitrogen adsorption methods. *J. Phys. Chem. C* **2011**, *115*, 10727– 929 10735.

930 (38) Gun'ko, V. M.; Kozynchenko, O. P.; Tennison, S. R.; Leboda, R.; 931 Skubiszewska-Zięba, J.; Mikhalovsky, S. V. Comparative study of nanopores in activated carbons by HRTEM and adsorption methods. 932 *Carbon* **2012**, *50*, 3146–3153. 933

(39) Gun'ko, V. M.; Turov, V. V.; Kozynchenko, O. P.; Nikolaev, V. 934 G.; Tennison, S. R.; Meikle, S. T.; Snezhkova, E. A.; Sidorenko, A. S.; 935 Ehrburger-Dolle, F.; Morfin, I.; Klymchuk, D. O.; Mikhalovsky, S. V. 936 Activation and structural and adsorption features of activated carbons 937 with highly developed micro-, meso- and macroporosity. *Adsorption* 938 **2011**, *17*, 453–460. 939

(40) Tascón, J. M. D., Ed.; Novel Carbon Adsorbents; Elsevier: 940 Amsterdam, 2012. 941

(41) Thommes, M.; Kaneko, K.; Neimark, A. V.; Olivier, J. P.; 942 Rodriguez-Reinoso, F.; Rouquérol, J.; Sing, K. S. W. Physisorption of 943 gases, with special reference to the evaluation of surface area and pore 944 size distribution (IUPAC Technical Report), Pure Appl. *Chem.* **2015**, 945 87, 1051–1069. 946

(42) Ravikovitch, P. I.; Neimark, A. V. Experimental confirmation of 947 different mechanisms of evaporation from ink-bottle type pores: 948 equilibrium, pore blocking, and cavitation. *Langmuir* **2002**, *18*, 9830–949 9837. 950

(43) Molina-Sabio, M.; González, M. T.; Rodríguez-Reinoso, F.; 951 Sepúlveda-Escribano, A. Effect of steam and carbon dioxide activation 952 in the micropore size distribution of activated carbon. *Carbon* **1996**, *34*, 953 505–509. 954

(44) Rodríguez-Reinoso, F.; Pastor, A. C.; Marsh, H.; Martínez. 955 Preparation of activated carbon cloths from viscous rayon. Part II: 956 physical activation processes. *Carbon* **2000**, *38*, 379–395. 957

(45) Kowalczyk, P.; Gauden, P. A.; Furmaniak, S.; Terzyk, A. P.; 958 Wiśniewski, M.; Ilnicka, A.; Łukaszewicz, J.; Burian, A.; Wloch, J.; 959 Neimark, A. V. Morphologically disordered pore model for character- 960 ization of micro-mesoporous carbons. *Carbon* **201**7, *111*, 358–370. 961

(46) Adem, Z.; Guenneau, F.; Springuel-Huet, M.-A.; Gédéon, A.; 962 Iapichella, J.; Cacciaguerra, T.; Galarneau, A. Diffusion properties of 963 hexane in pseudomorphic MCM-41 mesoporous silicas explored by 964 Pulsed Field Gradient NMR. *J. Phys. Chem. C* **2012**, *116*, 13749– 965 13759. 966

(47) Krutyeva, M.; Grinberg, F.; Furtado, F.; Galvosas, P.; Kärger, J.; 967 Silvestre-Albero, A.; Sepulveda-Escribano, A.; Silvestre-Albero, J.; 968 Rodríguez-Reinoso, F. Characterization of carbon materials with the 969 help of NMR methods. *Microporous Mesoporous Mater.* **2009**, *120*, 91–970 97. 971

(48) Kärger, J.; Ruthven, D. M.; Theodorou, D. N. Diffusion in 972 nanoporous materials; Wiley-VCH: Weinheim, 2012. 973

(49) Adem, Z.; Guenneau, F.; Springuel-Huet, M.-A.; Gédéon, A. 974 PFG NMR investigation of hydrocarbon diffusion in large NaX zeolite 975 crystals: Effect of internal field gradients on diffusion data. *Microporous* 976 *Mesoporous Mater.* **2008**, *114*, 337–342. 977

(50) Stallmach, F.; Gröger, S.; Künzel, V.; Kärger, J.; Yaghi, O. M.; 978 Hesse, M.; Müller, U. NMR studies on the diffusion of hydrocarbons on 979 the metal-organic framework material MOF-5. *Angew. Chem., Int. Ed.* 980 **2006**, 45, 2123–2126. 981

(51) Abdallah, W. Production, and characterisation of activated 982 carbon from sulphonated styrene divinylbenzene copolymer. The 983 Middle East Technical University: Ankara, 2004. https://etd.lib.metu. 984 edu.tr/upload/12605498/index.pdf (accessed 2023-12-21). 985